

## TITLE OF THE INVENTION

LASER PROCESS

## BACKGROUND OF THE INVENTION

### 1. Field of the Invention

The present invention relates to a highly reliable laser annealing process suited for use in mass production of semiconductor devices, which enables uniform annealing at high yield. More particularly, the present invention provides a laser annealing process of a deposited film whose crystallinity had been greatly impaired by the damage it had received through processes such as ion irradiation, ion implantation, and ion doping.

### 2. Prior Art

At present, methods of lowering of process temperatures in fabricating semiconductor devices are extensively studied. The reason for such an active research for low temperature processes owe partly to the need for fabricating semiconductor elements on an insulator substrate made of, e.g., glass. Laser annealing technology is regarded promising as the ultimate low temperature process.

However, conditions for laser annealing are not yet established because conventional laser annealing processes were each conducted independently under differing conditions which depend upon the apparatuses and the coating conditions chosen individually in each process. This has misled and has allowed many to think that the laser annealing technology fails to give results reliable and consistent enough to make the process practically feasible. An object of the present invention is to establish, for the first time, the conditions for a laser annealing process which yields highly reproducible results.

## SUMMARY OF THE INVENTION

In a process for fabricating a semiconductor device, a deposition film is considerably damaged by processing such as ion irradiation, ion implantation, and ion doping, and is thereby impaired in crystallinity as to yield an amorphous phase or a like state which is far from being called as a semiconductor. Accordingly, with an aim to use laser annealing in activating such damaged films, the present inventors have studied extensively how to optimize the conditions of laser annealing. During the study, it has been found that the optimum condition fluctuates not only by the energy control of the laser beam, but also by the impurities being incorporated in the film and by the number of pulse shots of the laser beam being applied thereto.

The deposited films to be activated by the process of the present invention are those containing, as the principal component, a Group IV element of the periodic table, e.g., silicon, germanium, an alloy of silicon and germanium, or a compound of the Group IV element such as silicon carbide. The deposited film has a thickness of 100Å to 10000Å. By taking the light transmission into consideration, it is well established that the laser annealing of such films can be favorably conducted by applying a laser beam in the short wavelength region, and specifically, one of 400 nm or shorter.

The process of the present invention comprises the step of:

irradiating laser pulses having a wavelength of 400nm or shorter and having a pulse width of 50nsec or less to a film comprising a Group IV element selected from the group consisting of carbon, silicon, germanium, tin and lead and having introduced thereinto an impurity ion,

wherein a transparent film having a thickness of 3 to 300nm is provided on said film comprising the Group IV element on

the way of said laser pulses to said film comprising the Group IV element, an energy density  $E$  of each of said laser pulses in unit of  $\text{mJ}/\text{cm}^2$  and the number  $N$  of said laser pulses satisfy relation  $\log_{10} N \leq -0.02(E-350)$ .

The laser pulses are emitted from a laser selected from the group consisting of a KrF excimer laser, an ArF excimer laser, a XeCl excimer laser and a XeF excimer laser. The introduction of the impurity ion is carried out by ion irradiation, ion implantation or ion doping. The film comprising the Group IV element is provided on an insulating substrate, and the insulating substrate is maintained at a temperature of room temperature to  $500^\circ\text{C}$  during the irradiating step.

It had been believed that the sheet resistance can be lowered by applying a laser beam having an energy density sufficiently high for activation. In the case of a film containing phosphorus as an impurity, this tendency can be certainly observed. However, in a film containing boron as an impurity, the film undergoes degradation by the irradiation of a laser of such a high energy density. Moreover, it had been taken for granted that the increase in pulsed shots reduces fluctuation in properties of the laser annealed films. However, this is not true because it was found that the morphology of the coating deteriorates with increasing number of shots to increase fluctuations in a microscopic level.

This can be explained by the growth of crystal nuclei within the coating due to a laser beam irradiation being applied repeatedly to the film. As a result, a grain size distribution within a size range of from  $0.1$  to  $1\ \mu\text{m}$  appears inside the coating which was previously composed of uniform sized grains. This phenomenon was particularly distinguished when a laser irradiation in the high energy region was applied.

It has been found that the deposited film (i.e. a

semiconductor film) must be coated with (covered by) a light-transmitting coating from 3 to 300 nm in thickness instead of being exposed to atmosphere. The light-transmitting coating is preferably made from silicon oxide or silicon nitride from the viewpoint that it should transmit laser beam. More preferably, a material mainly comprising silicon oxide is used because, in general, it also serves as the gate dielectric. Needless to say, the light-transmitting film may be doped with phosphorus or boron with an aim to passivate the mobile ions. If the film containing a Group IV element should not be coated with such a light-transmitting coating, it happens that the uniformity is disturbed in a more accelerated manner.

It has been found also, that a further smoother (uniform) coating can be obtained by applying pulsed laser beam under a condition set forth above and additionally satisfying the following relation:

$$\log_{10} N \leq A(E - B)$$

where,  $E$  ( $\text{mJ}/\text{cm}^2$ ) is the energy density of each of the irradiated laser pulses, and  $N$  (shots) is the number of shots of pulsed laser. The values for  $A$  and  $B$  are dependent on the impurities being incorporated in the coating. When phosphorus is present as the impurity,  $-0.02$  for  $A$  and  $350$  for  $B$  are chosen, and an  $A$  of  $-0.02$  and  $B$  of  $300$  are selected when boron is included as the impurity.

Similar effect can be attained by using a transparent substrate instead of the transparent film. That is, a laser process in accordance with the present invention comprises the steps of:

introducing an impurity into a semiconductor film provided on a transparent substrate; and

irradiating laser pulses having a wavelength of  $400\text{nm}$  or shorter and having a pulse width of  $50\text{ nsec}$  or less to said

semiconductor film through said transparent substrate,

wherein an energy density  $E$  of each of said laser pulses in unit of  $\text{mJ}/\text{cm}^2$  and the number  $N$  of said laser pulses satisfy relation  $\log_{10} N \leq -0.02(E-350)$ .

Fig.7(A) shows the introducing step, and Fig.7(B) shows the irradiating step. Reference numeral 71 designates the transparent substrate, and 72 designates the semiconductor film.

#### BRIEF DESCRIPTION OF THE DRAWINGS

FIG. 1 shows a schematic view of a laser annealing apparatus having used in the embodiments of the present invention;

FIG. 2 is a graph showing the relationship between the sheet resistance of a silicon film (phosphorus-doped, N-type) obtained by laser annealing according to an embodiment of the present invention and the applied laser energy density, while changing the repetition times of pulse shots;

FIG. 3 is a graph showing the relationship between the sheet resistance of a silicon film (phosphorus- and boron-doped, P-type) obtained by laser annealing according to an embodiment of the present invention and the applied laser energy density, while changing the repetition times of pulse shots;

FIG. 4 is a graph showing the relation between the morphology of the silicon film obtained in an embodiment of the present invention and the applied laser energy density and the repetition times of the pulse shots;

Fig.5 shows a concept of an optical system of the laser annealing apparatus having used in the embodiments of the present invention;

Fig.6 shows a laser annealing process in accordance with the present invention; and

Fig.7 shows another laser annealing process in accordance with the present invention.

## DETAILED DESCRIPTION OF THE INVENTION

The present invention is illustrated in greater detail referring to a non-limiting example below. It should be understood, however, that the present invention is not to be construed as being limited thereto.

### EXAMPLE

In this EXAMPLE, an impurity is introduced into a film comprising a Group IV element for imparting one of N-type conductivity and P-type conductivity thereto, and another impurity is introduced into a portion of the film with a mask for imparting the other one of the N-type conductivity and P-type conductivity to said portion. In FIG. 1 is shown schematically a laser annealing apparatus having used in the present example. A laser beam is generated in a generator 2, amplified in an amplifier 3 after traveling through full reflection mirrors 5 and 6, and then introduced in an optical system 4 after passing through full reflection mirrors 7 and 8. The initial laser beam has a rectangular beam area of about  $3 \times 2 \text{ cm}^2$ , but is processed into a long beam having a length of from about 10 to 30 cm and a width of from about 0.1 to 1 cm by the optical system 4. The maximum energy of the laser having passed through this optical system was 1,000 mJ/shot.

An optical path in the optical system 4 is illustrated in Fig.5. A laser light incident on the optical system 4 passes through a cylindrical concave lens A, a cylindrical convex lens B, a fly-eye lens C provided in a lateral direction and a fly-eye lens D provided in a vertical direction. The laser light is changed from an initial gauss distribution to a rectangular distribution by virtue of the fly-eye lenses C and D. Further, the laser light passes through a cylindrical convex lenses E and

F and is reflected on a mirror G (a mirror 9 in Fig.1) and is focused on the specimen by a cylindrical lens H.

In this EXAMPLE, distances  $X_1$  and  $X_2$  indicated in Fig.5 are fixed, and a distance  $X_3$  between a virtual focus I (which is generated by the difference between curved surfaces of the fly-eye lenses) and the mirror G, distances  $X_4$  and  $X_5$  are varied to adjust a magnification M and a focal length F. That is,

$$M=(X_3+X_4)/X_5$$

$$1/F=1/(X_3+X_4)+1/X_5.$$

In this EXAMPLE, a total length  $X_6$  of the optical path is about 1.3m.

The initial beam is modified into a long-shaped one as above to improve processability thereof. More specifically, the rectangular beam which is irradiated onto a specimen 11 through the full reflection mirror 9 after departing the optical system 4 has a longer width as compared with that of the specimen that, as a consequence, the specimen need to be moved only along one direction. Accordingly, the stage on which the specimen is mounted and the driving apparatus 10 can be made simple structured that the maintenance operation therefor can be easily conducted. Furthermore, the alignment operation at setting the specimen can also be greatly simplified.

If a beam having a square cross section were to be employed, on the other hand, it becomes impossible to cover the entire substrate with a single beam. Accordingly, the specimen should be moved two dimensionally along two directions. In such circumstances, however, the driving apparatus of the stage becomes complicated and the alignment also must be done in a two dimensional manner that it involves much difficulty. When the alignment is done manually, in particular, a considerable time is consumed for this step to greatly reduce the productivity of the entire process. Furthermore, those apparatuses must be fixed on a

stable table 1 such as a vibration proof table.

The specimen used in the example were various types of glass substrates (e.g., a Corning #7059 glass substrate) 100 mm in length and from 100 to 300 mm in width. A KrF laser emitting light at a wavelength of 248 nm and at a pulse width of 50nsec or less, e.g. 30 nsec, was used in the process.

A 100 nm thick amorphous silicon film was deposited on a glass substrate 61 by plasma assisted CVD (chemical vapor deposition) process. The resulting film was annealed at 600°C for 48 hours to obtain a crystallized film, and was patterned to make island-like portions 62 and 63 (Fig.6(A)). Furthermore, a 70 nm thick silicon oxide film (a light-transmitting coating) 64 was deposited thereon by sputtering and the entire surface of the substrate was doped with phosphorus. A so-called ion doping process (Fig.6(B)) was employed in this step using phosphine ( $\text{PH}_3$ ) as the plasma source and an accelerating voltage of 80 kV. Furthermore, a part of the substrate was masked 65 to implant boron by ion doping process (Fig.6(C)). Diborane ( $\text{B}_2\text{H}_6$ ) was used as the plasma source in this step while accelerating at a voltage of 65 kV. More specifically, phosphorus was implanted (introduced) into the masked portions through the light-transmitting coating to obtain portion having rendered N-type conductive, while both phosphorus and boron were implanted (introduced) into the unmasked portions through the light-transmitting coating to result in a portion having rendered P-type conductive.

Then, laser beam was irradiated to the island-like portions (semiconductor film) while varying the energy density and the number of pulse shots to effect laser activation. The sheet resistance was measured accordingly and the morphology of the crystallites constituting the coating was observed through an optical microscope. The results are summarized in FIGs. 2 to 4.



FIG. 2 shows a graph which relates the sheet resistance of a silicon film having doped with phosphorus ions with the energy density of the laser beam while also changing the repetition of the pulse shots. Phosphorus was incorporated into the silicon film at a dose of  $2 \times 10^{15} \text{ cm}^{-2}$ . With a laser being operated at an energy density of  $200 \text{ mJ/cm}^2$  or less, a large number of shots were necessary to activate the sheet, yet with a poor result yielding a high sheet resistance of about  $10 \text{ k}\Omega/\text{sq}$ . However, with a laser beam having an energy density of  $200 \text{ mJ/cm}^2$  or higher, a sufficient activation was realized with a laser operation of from 1 to 10 shots.

In FIG. 3 is shown the results for laser activating a silicon film doped with boron ions at a dose of  $4 \times 10^{15} \text{ cm}^{-2}$ . In this case again, activation could be conducted only insufficiently with an energy density of  $200 \text{ mJ/cm}^2$  or lower that a large number of pulse shots was required for sufficient activation. With a laser beam operated at an energy density of from 200 to  $300 \text{ mJ/cm}^2$ , a sufficiently low sheet resistance was obtained with 1 to 10 shots. However, with laser being operated at an energy density of  $300 \text{ mJ/cm}^2$  or higher, on the other hand, the sheet resistance was reversely elevated. In particular, contrary to the case of activating with a laser beam energy density of  $200 \text{ mJ/cm}^2$  or lower, the sheet resistance was elevated with increasing repetition of pulse shots. This phenomenon can be explained by the growth of grain boundary due to the impaired homogeneity of the film which had resulted by applying laser irradiation for too many shots.

In a practical process, the laser annealing is applied simultaneously to both P- and N-type regions as shown in Fig.6(D). This signifies that a laser beam being irradiated at an energy density of  $350 \text{ mJ/cm}^2$  sufficiently activates the N-type region while impairing the properties of the P-type region.

Accordingly, in the process according to the present example, it is preferred that the laser beam is operated in an energy density range of from 200 to 300 mJ/cm<sup>2</sup>, and more preferably, in a range of from 250 to 300 mJ/cm<sup>2</sup>. The [pulse repetition] is preferably in the range of from 1 to 100 pulses.

As described in the foregoing, the morphology of the deposited film is considerably influenced by laser annealing. In fact, the number of pulse shots can be related to the laser beam energy density and the film morphology as illustrated in FIG. 4. In FIG. 4, the term "Annealing Pulse" signifies the number of laser beam pulse shots. The solid circle in the figure represents the point at which a change in surface morphology was observed on a phosphorus-doped silicon, and the open circle represents the same on a boron-doped silicon. The upper region on the right hand side of the figure corresponds to a condition which yields poor morphology on the surface (rough surface), and the lower region on the left hand side of the figure corresponds to that which yields favorable morphology on the surface (smooth surface). It can be seen from the results that the phosphorus-doped silicon has a strong resistance against laser irradiation. Accordingly, the condition for conducting laser annealing without impairing the surface morphology can be read to be such which satisfies the relation :

$$\log_{10} N \leq A(E - B),$$

where, E (mJ/cm<sup>2</sup>) is the energy density of the irradiated laser beam, and N (shots) is the number of shots of pulsed laser. The values for A and B are A=-0.02 and B=350 in the case phosphorus is incorporated as the impurity, and are A=-0.02 and B= 300 when boron is included as the impurity.

When the morphology of the deposited film is considerably impaired, the characteristic values show large scattering due to the serious drop which occurs locally in the properties of

silicon. In fact, a scattering in sheet resistance as high as 20% or even more was observed on a silicon film having a defective morphology (a rough surface). This scattering can be removed by satisfying the conditions above and by setting the laser energy density at a pertinent value.

For instance, when a laser energy density is set at 250 mJ/cm<sup>2</sup>, the pulsed laser beam is shot at a frequency of 10 times or less. If the energy density is elevated to 280 mJ/cm<sup>2</sup>, the laser beam is preferably shot at a frequency of from 1 to 3 times. By conducting laser annealing under such conditions, the sheet resistance could be controlled within a fluctuation of 10% or less.

According to the present invention, a highly reliable semiconductor film having low fluctuation in properties was obtained by setting the optimal conditions for laser annealing as described in the foregoing. It can be seen therefore that the process according to the present invention is beneficial to the semiconductor industry.

While the invention has been described in detail and with reference to specific embodiments thereof, it will be apparent to one skilled in the art that various changes and modifications can be made therein without departing from the spirit and scope thereof.